

## Electronic and magnetic structure of $\text{Fe}_3\text{S}_4$ : GGA+U investigation

A. J. Devey,<sup>\*</sup> R. Grau-Crespo, and N. H. de Leeuw

*Department of Chemistry, University College, University of London, 20 Gordon Street, London WC1H 0AJ, United Kingdom*

(Received 10 February 2009; revised manuscript received 18 March 2009; published 26 May 2009)

The electronic and magnetic behavior of the iron sulphide mineral greigite ( $\text{Fe}_3\text{S}_4$ ) is studied using *ab initio* density-functional theory in the generalized gradient approximation (GGA) with the on-site Hubbard  $U_{\text{eff}}$  parameter (GGA+U). The effect of the Hubbard correction is investigated and is found to be a necessary requirement for the accurate description of both the unit cell structure and the magnetic moment. A ferrimagnetic normal-spinel structure is found when  $U_{\text{eff}}=0$  eV, while for all values of  $U_{\text{eff}}>0$  eV an inverse spinel structure is predicted, in agreement with experiment. For low values of  $U_{\text{eff}}$  ( $0 < U_{\text{eff}} < 4$  eV) the predicted electronic structure corresponds to that of a semimetal, with semimetallicity arising from electron hopping between ferric and ferrous Fe on octahedral sites. For values of  $U_{\text{eff}} \geq 4$  eV the S atoms are found to oxidize the ferrous octahedral sites Fe to the ferric state. To determine whether GGA+U predicts a stable monoclinic form of greigite arising from a Verwey-type low-temperature transition, analogous to that seen in magnetite, a monoclinic form of greigite is postulated. It is found that such a phase is stable, with an electronic band-gap opening up for values of  $U_{\text{eff}} \geq 2$  eV, but is energetically unfavorable when compared with the spinel phase for all  $U_{\text{eff}}$  values tested. It is argued that an accurate description of all the properties of greigite requires a  $U_{\text{eff}}$  value of approximately 1 eV.

DOI: [10.1103/PhysRevB.79.195126](https://doi.org/10.1103/PhysRevB.79.195126)

PACS number(s): 71.20.-b, 71.15.Mb, 71.27.+a, 71.28.+d

### I. INTRODUCTION

Greigite ( $\text{Fe}_3\text{S}_4$ ) is the sulphide counterpart of the well-known iron oxide spinel magnetite.<sup>1</sup> First defined as a mineral by Skinner *et al.*<sup>2</sup> from a Californian lacustrine sediment sequence, it is now considered a common magnetic material<sup>3</sup> and has been found in many natural environments with ages up to several million years old.<sup>4</sup> Research has suggested that greigite may have played an important role as a catalyst in the development of protometabolism, primarily due to its similarity to the cubane cluster structure  $\text{Fe}_4\text{S}_4$ ,<sup>5</sup> which is still found as the active sites in many enzymes.<sup>6</sup> In addition, greigite has been discovered in the scales of a deep sea hydrothermal vent gastropod,<sup>7</sup> it is an important paleomagnetic material<sup>1</sup> and is widespread in magnetostatic bacteria.<sup>8</sup> Greigite is believed to form from mackinawite ( $\text{Fe}^{2+}\text{S}^{2-}$ ) via oxidation of two-thirds of the  $\text{Fe}^{2+}$  cations, together with rearrangement about the cubic close-packed S anion sublattice<sup>9</sup> which in turn undergoes a small volume reduction of around 12%.<sup>10</sup> Recently, both one-dimensional rods<sup>11</sup> and two-dimensional nanosheets<sup>12</sup> of greigite have been synthesized, opening up the possibility of research into the magnetic properties of low-dimensional iron sulphide structures.

Greigite has a cubic unit cell containing 8 iron cations in tetrahedral coordination (hereafter referred to as A sites) and 16 iron cations in octahedral coordination (B sites) with 32 sulfur anions (Fig. 1).<sup>13</sup> The magnetic moments on the tetrahedral and octahedral Fe sublattices are aligned in an antiparallel fashion, rendering greigite ferrimagnetic.<sup>14</sup>

Measurements of the magnetization via high-field experiments at room temperature give a saturation magnetization for greigite of  $3.13 \mu_B$  per formula unit (f.u.), which at 5 K increases to  $3.35 \mu_B/\text{f.u.}$  due to decreased thermal excitation.<sup>15</sup> This recent result is in contrast to the previously determined experimental value of  $2.2 (\pm 0.3) \mu_B/\text{f.u.}$ ,<sup>14</sup> presumed to be due to the presence of impurities, vacancies or a

combination of both in the greigite samples tested. In any case, the magnetization of greigite is below the value of  $\sim 4.0 \mu_B/\text{f.u.}$  observed in magnetite,<sup>16</sup> which is the value expected from a purely ionic model ( $4 \mu_B$  on  $\text{Fe}^{2+}$ ;  $5 \mu_B$  on  $\text{Fe}^{3+}$ ). Mössbauer measurements performed on both natural and synthetic greigite samples<sup>14,17-20</sup> have demonstrated that the cations on the tetrahedral and octahedral sublattices order with an inverse spinel structure,  $(\text{Fe}^{3+})_A(\text{Fe}^{2+}\text{Fe}^{3+})_B\text{S}$ , down to a temperature of at least 4.2 K. Recent measurements have suggested an estimate for the  $J_{AB}$  exchange constant between tetrahedral and octahedral sublattices in greigite of  $\sim 1.03$  meV,<sup>15</sup> lower than that calculated for magnetite (2.88 meV),<sup>21</sup> indicating a lower level of magnetic coupling in the sulfide.

Spender *et al.*<sup>14</sup> carried out measurements of the conductivity of greigite and found the presence of delocalized

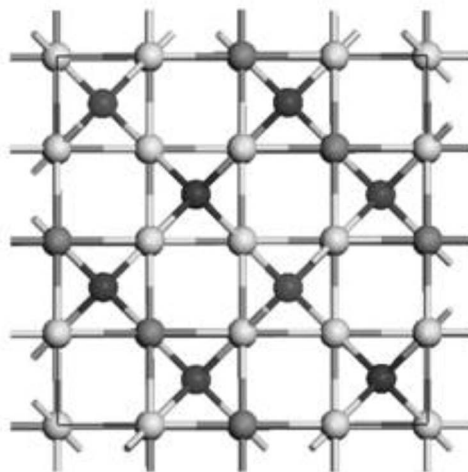


FIG. 1. Structure of greigite viewed along the  $\langle 100 \rangle$  axis. The S atoms are shown in white, the tetrahedral Fe in dark gray, and the octahedral Fe in light gray.

charge carriers in the compound, which they attributed to electron hopping between ferrous and ferric Fe on octahedral sites, leading to a semimetallic nature. Regarding the electronic structure of greigite, two band schemes have been proposed: The first suggests that greigite possesses an average of ferric ( $d^5$ ) and ferrous ( $d^6$ ) iron on  $B$  sites, similar to that found in magnetite;<sup>22</sup> while the second assumes that ferric iron on the  $B$  sites is reduced by the sulfur anions and thus only ferrous iron is present in octahedral coordination.

Experimental low-temperature magnetization measurements have failed to reach a consensus regarding the existence of a Verwey-type transition in greigite at temperatures between 4.2 and 300 K, and it is not clear whether a transition would be discernable using such methods. While some studies have not detected a transition,<sup>23–25</sup> it has been pointed out that the greigite samples tested may not be sufficiently stoichiometric for the transition to be apparent, possibly due to the presence of  $B$  site vacancies. A maximum in remnant saturation magnetization measured at 10 K for a variety of both synthetic and natural greigite samples has been suggested as evidence of a transition, although other mechanisms may be responsible which are unconnected with any transition.<sup>4</sup>

A number of questions remain unanswered in relation to the electronic structure and magnetic behavior of greigite that are well suited to an investigation by modern *ab initio* techniques. The main aim of this work is to investigate the electronic structure of greigite. Similar to the findings of studies into magnetite,<sup>26,27</sup> the importance of the on-site Fe electronic correlation represented by the  $U_{\text{eff}}$  parameter in the description of greigite will be investigated. Finally, we explore the possibility of any low-temperature Verwey-type transition in greigite.

## II. METHODOLOGY

The greigite structure is modeled using the Vienna *ab initio* simulation program (VASP),<sup>28–31</sup> which employs spin-polarized density-functional theory (DFT) with a basis set constructed from plane waves. The theory and application of the plane-wave DFT methodology have been described extensively elsewhere;<sup>32</sup> it is well established and has been applied to a wide range of materials including transition metal sulphides,<sup>33–35</sup> oxides<sup>36–38</sup> and spinels.<sup>27,39</sup> All calculations are performed within the generalized gradient approximation (GGA), using the exchange-correlation functional developed by Perdew *et al.*<sup>40</sup> and the spin interpolation formula of Vosko *et al.*<sup>41</sup> The interaction between the valence electrons and the core is described with the projector augmented wave (PAW) method<sup>42</sup> in the implementation of Kresse and Joubert.<sup>43</sup> The core levels, which are kept frozen during the calculations, consisted of orbitals up to, and including, the  $3p$  levels for Fe and the  $2p$  level for S. Geometry optimization of the 56-atom cubic unit cell is performed with an energy cutoff of 600 eV and a Monkhorst-Pack (MP)<sup>44</sup> grid of  $4 \times 4 \times 4$ , while the 56-atom monoclinic unit cell used the same cutoff energy and a MP grid of  $4 \times 4 \times 2$ . This high value for the cutoff energy ensured that no Pulay stresses occurred within the cell during relaxations. In order to im-

prove the convergence of the Brillouin-zone integrations, the partial occupancies were determined using Gaussian smearing, with a set width for all calculations of 0.02 eV. These smearing techniques can be considered in the form of a finite-temperature DFT,<sup>45</sup> where the variational quantity is the electronic free energy. The optimization of the structures was conducted via a conjugate gradients technique, which uses the total energy and the Hellmann-Feynman forces on the atoms. Spin-orbit coupling was not taken into account.

It has been pointed out in several studies of transition metal compounds<sup>46–48</sup> that the GGA methodology often provides an unsatisfactory description of such highly correlated materials. Thus we have also performed calculations using the so-called GGA+ $U$  method, in the formulation of Lichtenstein<sup>49</sup> and later Dudarev,<sup>50</sup> where a single parameter,  $U_{\text{eff}}$ , determines an orbital-dependent correction to the GGA energy,

$$E_{\text{GGA}+U} = E_{\text{GGA}} + U_{\text{eff}} \sum_{\sigma} \text{Tr}[\rho^{\sigma} - \rho^{\sigma} \rho^{\sigma}], \quad (1)$$

where  $\rho^{\sigma}$  is the on-site density matrix with spin component  $\sigma$ . The parameter  $U_{\text{eff}}$  is generally expressed as the difference between two parameters, the Hubbard  $U$ , which is the Coulomb-energetic cost to place two electrons at the same site, and an approximation of Hund's exchange parameter  $J$ , which is almost constant at  $\sim 1$  eV.<sup>51</sup> The GGA+ $U$  correction alters the one-electron potential locally for the specified orbitals, here the Fe  $d$  orbitals, reducing the hybridization with the S ligands. The  $U_{\text{eff}}=0$  case represents the GGA limit. Details of the implementation of the GGA+ $U$  method in the VASP code can be found in the work of Rohrbach *et al.*,<sup>34</sup> where it was also shown that although the  $U_{\text{eff}}$  parameter introduces a form of semiempiricism into the calculations, the Hubbard correction improves the description of many transition metal sulphides. It was also instrumental in the description of the low-temperature magnetite structure.<sup>26</sup> The  $U_{\text{eff}}$  method is chosen over other exchange functional methods such as hybrid functionals due to the large system size. A useful comparison of GGA+ $U$  with the hybrid functional B3LYP for the antiferromagnetic material FeSbO<sub>4</sub> is given in Ref. 38. Work is ongoing regarding improvements in both the GGA<sup>52</sup> and hybrid methods.<sup>53</sup>

The greigite spinel structure determined by Uda<sup>13</sup> is used as the starting geometric arrangement, and a full relaxation of the unit cell volume, shape, and internal atomic coordinates is undertaken, followed by a second relaxation solely of the internal coordinates in order to ensure full relaxation of the structure. Finally, a single-point calculation of the electronic structure is completed, using a tetrahedral smearing method with Blochl corrections in order to obtain the electronic ground state and the electronic density of states.

## III. RESULTS

### A. Spinel structure

The 56-atom cubic unit cell of greigite is modeled using GGA+ $U$  calculations for a range of  $U_{\text{eff}}$  values and three different initial magnetic arrangements: nonmagnetic (zero

initial magnetic moments on all Fe sites); inverse spinel (initial magnetic moments of  $5 \mu_B$  on each of the tetrahedral Fe sites and  $4.5 \mu_B$  on each of the 16 octahedral sites); and normal-spinel (initial magnetic moments of  $4 \mu_B$  on each of the tetrahedral sites and  $5 \mu_B$  on each of the octahedral sites). For  $U_{\text{eff}}=0$  eV, the VASP calculation of the greigite spinel structure gives a cubic structure with lattice parameter  $a=9.48$  Å. It was found that the resulting magnetic structure is always ferrimagnetic with a magnetization per formula unit of  $2.08 \mu_B$ , regardless of the initial arrangement of the magnetic moments. These results show only moderate agreement with the experimentally determined values for greigite;  $a=9.88$  Å and  $3.35 \mu_B$  for the lattice parameters and magnetic moment per formula unit respectively.<sup>13,15</sup>

In order to determine whether the introduction of the Hubbard parameter improves the DFT description of greigite, suitable values for the  $U_{\text{eff}}$  parameter are applied. Identical cell relaxations as for the  $U_{\text{eff}}=0$  eV case are undertaken. The  $U_{\text{eff}}$  values chosen vary from 0.5 to 5 eV, in steps of 0.5 eV.

The cubic unit cell is accurately reproduced for all cases, with  $a=b=c$  and angles at  $90^\circ$ . The initial magnetic moment was found to be inconsequential, with all three cases converging to a stable ferrimagnetic state for each  $U_{\text{eff}}$  value tested. The calculated lattice parameters for each value of  $U_{\text{eff}}$  are shown in Fig. 2(a). The introduction of the  $U_{\text{eff}}$  value has a significant effect upon the lattice parameters, compensating for the overbinding seen in the  $U_{\text{eff}}=0$  eV case and yielding the experimentally determined value for the lattice parameter at a  $U_{\text{eff}}$  of between 1 and 1.5 eV. It is worth noting that this value is similar to the value of  $U_{\text{eff}}$  found to give an accurate description of troilite (FeS).<sup>34</sup> The experimental value for the  $S u$  parameter of 0.2505, which represents the first internal S coordinate within the unit cell, is well reproduced for all  $U_{\text{eff}}$  values.

A plot of the magnetization per formula unit versus  $U_{\text{eff}}$  for the spinel structure is given in Fig. 2(b), and a plot of the magnitude of each individual magnetic moment from each sublattice is shown in Fig. 2(c). The magnetic moment on each site is found using a Bader analysis, where the electron spin density associated with each atom is integrated over the Bader volume of the atom in question.<sup>54</sup> The use of Bader analysis is justified by the fact that the effective radius of an ion changes with the oxidation state, and therefore it is not correct to perform the integration around a sphere of constant radius, when considering mixed-valence systems such as greigite.

The magnetic moments on the Fe atoms when  $U_{\text{eff}}=0$  eV are much lower than the values in magnetite<sup>16</sup> or than would occur in the purely ionic case of integer unpaired electrons. This calculation gives a net magnetic moment per formula unit of 60% of the experimentally determined value, which arises from an overestimation of the covalency of the Fe-S bond by the pure GGA. Introducing the  $U_{\text{eff}}$  parameter leads to an increase in the total magnetic moment, caused by an underlying increase in the magnetic moments on both Fe<sub>A</sub> and Fe<sub>B</sub> sites. As the  $U_{\text{eff}}$  value is increased to 2 eV, the total magnetic moment reaches a maximum of  $3.9 \mu_B/\text{f.u.}$ , close to the value of  $4 \mu_B/\text{f.u.}$  predicted for a purely ionic model. The magnetic moment on every Fe atom reaches a maximum

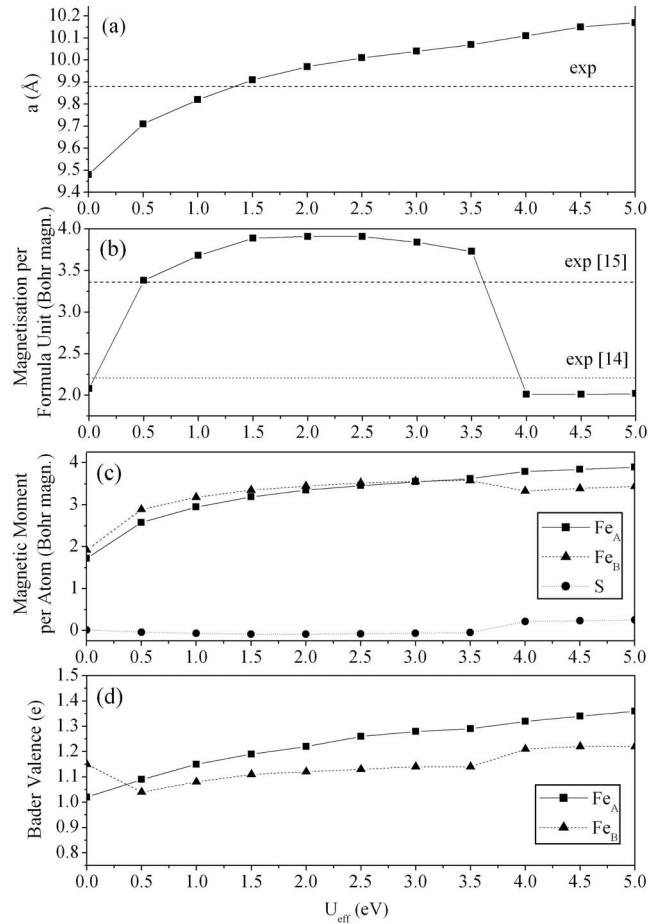


FIG. 2. (a) Calculated lattice constant for the cubic spinel structure of greigite as a function of the effective Hubbard parameter  $U_{\text{eff}}$ . The dashed line shows the experimentally determined value for  $a$  (Ref. 13). (b) Magnetization per formula unit versus  $U_{\text{eff}}$  for the spinel structure of greigite. The dashed line gives the experimental value of Chang *et al.* (Ref. 15) and the dotted line the value of Spender *et al.* (Ref. 14) (c) Magnitude of greigite of the magnetic moments on each sublattice of greigite versus  $U_{\text{eff}}$ . (d) Fe<sub>A</sub> and Fe<sub>B</sub> atomic Bader populations versus  $U_{\text{eff}}$  value.

of  $3.6 \mu_B/\text{f.u.}$  at  $U_{\text{eff}}=3.5$  eV, and at this point the magnetic moments of both the tetrahedral and octahedral Fe sites are identical. From  $U_{\text{eff}}=4$  eV upwards each S atom develops a nonzero magnetic moment of magnitude  $0.2 \mu_B$ , parallel in direction to that of the Fe<sub>A</sub> atoms. In addition a difference of  $0.45 \mu_B$  develops between the magnetic moments of the Fe atoms on the A and B sites, with a value on the tetrahedral sites of around  $3.8 \mu_B$  and the octahedral sites of  $3.3 \mu_B$ . These two factors act to reduce the net total magnetic moment to a value of around  $2.0 \mu_B/\text{f.u.}$  for  $U_{\text{eff}} \geq 4$  eV. It is noted that the experimentally determined value for the magnetic moment of  $3.35 \mu_B/\text{f.u.}$  is achieved at around  $U_{\text{eff}}=0.5$  or  $3.7$  eV.

The variation in the Bader charge populations associated with the Fe<sub>A</sub>, Fe<sub>B</sub>, and S sites with the  $U_{\text{eff}}$  parameter is shown in Fig. 2(d). For the case of  $U_{\text{eff}}=0$  eV, these populations indicate that there is a greater number of electrons on the Fe<sub>A</sub> atoms than the Fe<sub>B</sub>, corresponding to the electronic structure of a normal spinel (based on the assumption that



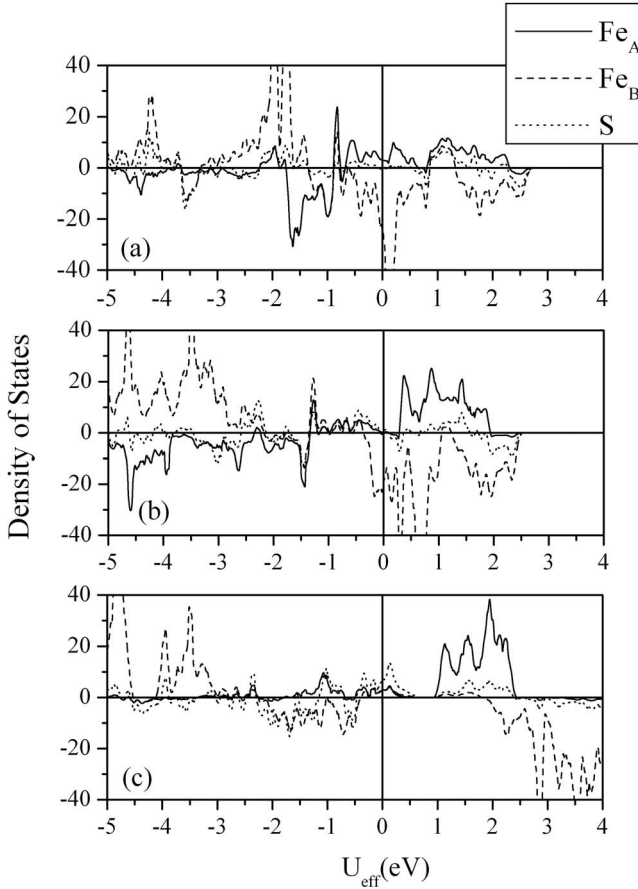


FIG. 3. Electronic DOS for the spinel form of greigite, for (a)  $U_{\text{eff}}=0$  eV, (b)  $U_{\text{eff}}=1$  eV, and (c)  $U_{\text{eff}}=5$  eV. Contributions from each sublattice are plotted.

the greater number of electrons signifies a valence of  $\text{Fe}^{2+}$  and lesser number of electrons denotes a share of  $\text{Fe}^{2+}$  and  $\text{Fe}^{3+}$ ). The introduction of the  $U_{\text{eff}}$  parameter causes the valence of the  $\text{Fe}_A$  sites to increase relative to that of the  $\text{Fe}_B$  sites, with the effect that there is a crossover of the valences of these sublattices. Thus more valence electrons are associated with the  $B$  sites than the  $A$  sites, and this relation remains for all non-zero  $U_{\text{eff}}$  values tested. Thus it can be inferred that the introduction of the  $U_{\text{eff}}$  parameter has the effect of changing the electronic structure from that of the normal spinel to that of the inverse spinel. Since it is the inverse spinel that is observed experimentally, this highlights the importance of the  $U_{\text{eff}}$  parameter in the description of greigite. Between the values of  $U_{\text{eff}}=3.5$  and  $U_{\text{eff}}=4$  eV, the  $\text{Fe}_B$  atoms experience a sudden reduction in Bader charge valence, and the electrons are transferred to the  $S$  atoms. This scenario corresponds to the second band picture suggested by Spender *et al.*<sup>14</sup> where the  $\text{Fe}_B$  are reduced by  $S$ .

Figures 3(a)–3(c) show the electronic density of states (DOS) of the spinel form of greigite for the  $U_{\text{eff}}=0$ , 1, and 5 eV cases, respectively. The DOS for  $U_{\text{eff}}=0$  eV shows that the available states at the Fermi level arise from both the spin-down  $\text{Fe}_B$  sites and the spin-up  $\text{Fe}_A$  sites, a situation not seen for the  $U_{\text{eff}}=0$  eV magnetite spinel previously modeled by Piekarczyk *et al.*<sup>36</sup> The strong effect of the  $U_{\text{eff}}$  parameter upon the  $\text{Fe}_A$  bands around the Fermi level is clearly seen in

the DOS for  $U_{\text{eff}}=1$  eV case [Fig. 3(b)]. A gap of 0.3 eV opens between the  $e$  and  $t_2$   $3d$  energy levels of the  $\text{Fe}_A$  band, while the  $\text{Fe}_B$  band is largely unaffected compared to  $U_{\text{eff}}=0$  eV. This leads to a semimetallic band structure for greigite, with the spin-down  $\text{Fe}_B$  minority band providing states at the Fermi energy and a band gap in the  $\text{Fe}_A$  spin-up band. Figure 3(c) shows the DOS for  $U_{\text{eff}}=5$  eV, which shows that for large  $U_{\text{eff}}$  values a splitting of the spin-down  $\text{Fe}_B$  band occurs, clearly revealing the  $t_{2g}$  and  $e_g$  energy levels of the  $3d$  orbital. The  $\text{Fe}_B$   $d$ -orbital spin-down band no longer occupies the energies around the Fermi level, and the semimetallic behavior disappears. The majority of states are provided by holes in the spin-up  $S$  band.

## B. Monoclinic structure

Experimental investigations at low temperatures have so far been unable to provide a definitive answer as to whether a Verwey transition occurs in the greigite structure. Studies using a similar theoretical framework have proved highly successful in the description of the low-temperature monoclinic form of magnetite.<sup>36</sup> By analogy, a hypothetical monoclinic form of greigite is postulated, and GGA+ $U$  is used to determine its energetic stability compared to the spinel structure. The following calculations are based upon the low-temperature monoclinic structure of magnetite determined by Wright *et al.*,<sup>55</sup> but with the lattice parameters scaled up to account for the larger anion radius in the sulphide compared to the oxide. The scaling constant for each orthogonal lattice direction is given by the ratio of the spinel greigite lattice constant  $a_{\text{grei}}$  to that of the spinel magnetite structure  $a_{\text{mag}}$ , where  $a_{\text{grei}}/a_{\text{mag}}=9.88/8.39=1.18$ . Scaling each monoclinic magnetite lattice parameter by this factor gives estimates of  $a=6.99$  Å,  $b=6.98$  Å, and  $c=19.75$  Å for the hypothetical monoclinic greigite structure. The same simulations as for the spinel structure are then repeated for a range of  $U_{\text{eff}}$  values from 0 to 5 eV, in steps of 1 eV. All relaxations yield stable monoclinic structures, with lattice parameters given in Table I.

The total magnetization per formula unit for the 56-atom monoclinic unit cell of greigite is shown in Fig. 4. For low values of  $U_{\text{eff}} (<4$  eV), monoclinic structures with net magnetic moments of 1.7 to 2  $\mu_B/\text{f.u.}$  are found, indicating that if a transition to this structure did occur it would be accompanied by a large, observable reduction in the magnetic moment. It is noted that at low  $U_{\text{eff}}$  values, there is a splitting of the symmetry of the  $\text{Fe}_B$  sites, to the degree that the magnetic moment of half the sites is 60% greater than that of the other half. For values of  $U_{\text{eff}} \geq 4$  eV, the electronic structure becomes even more complex, with four identifiable groups of four  $\text{Fe}_B$  sites, in a manner similar to the charge disproportionation seen in the low-temperature phase of magnetite. The values of the band gap for each  $U_{\text{eff}}$  value are listed in Table I. The DOSs for the monoclinic form of greigite for  $U_{\text{eff}}$  values of 0, 1, and 5 eV are shown in Figs. 5(a)–5(c), respectively. For  $U_{\text{eff}}=1$  eV, both  $\text{Fe}_A$  and  $\text{Fe}_B$  sublattices provide available states at the Fermi level. As  $U_{\text{eff}}$  is increased to 1 eV a band gap opens in the  $\text{Fe}_B$  band. For  $U_{\text{eff}}=5$  eV a band gap for both  $\text{Fe}_A$  and  $\text{Fe}_B$  sublattices opens

TABLE I. Calculated lattice parameters and band-gap width for the theoretical monoclinic form of greigite for a range of  $U_{\text{eff}}$ . The difference in the internal energies  $\Delta E$  of the 56-atom unit cells of the spinel and monoclinic forms of greigite over the range of  $U_{\text{eff}}$  values modeled is also presented.

$U_{\text{eff}}$ (eV)	$a$ (Å)	$b$ (Å)	$c$ (Å)	Band gap (eV)	$\Delta E$ (eV)
0	6.57	6.75	18.99	0.00	1.15
1	6.79	6.87	19.52	0.00	2.69
2	6.93	6.96	19.84	0.06	3.69
3	6.96	6.99	20.02	0.16	4.55
4	7.09	7.16	20.21	0.14	2.71
5	7.29	7.24	20.61	0.29	0.92

and the structure becomes insulating, similar to that observed in simulations of monoclinic magnetite.<sup>36</sup>

The difference in the internal energies of the spinel and monoclinic structures for the range of  $U_{\text{eff}}$  values is given in Table I. It is clear from the calculated total energies of the two structures that the monoclinic form is only metastable with respect to the spinel. The precise energy difference between the two depends on the values of  $U_{\text{eff}}$ , but for all values the spinel structure is energetically favored.

#### IV. CONCLUSION

In this work we have used the GGA+U approach, where  $U$  is the on-site Hubbard  $U_{\text{eff}}$  parameter, to investigate the energetic, electronic and magnetic properties of a spinel and monoclinic structure of Fe<sub>3</sub>S<sub>4</sub>. Simulations of the spinel structure over the range  $0 \leq U_{\text{eff}} \leq 5$  eV result in stable ferromagnetic structures, with Fe atoms on the tetrahedral and octahedral sublattices aligned in an antiparallel manner in accord with published experimental findings. GGA in the absence of any  $U_{\text{eff}}$  correction leads to a large underestimation of the lattice parameter and the magnetic moment, as well as an electronic arrangement whereby the Bader charges of the tetrahedral and octahedral Fe sites form a normal-spinel arrangement. These errors are thought to arise from the GGA failing to take into account the electron correlation associated with the Fe atoms. The experimentally determined inverse spinel structure is correctly simulated upon the intro-

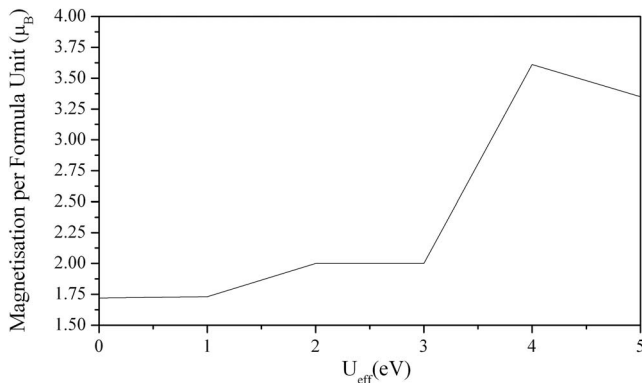


FIG. 4. Total magnetization per formula unit versus  $U_{\text{eff}}$  for the monoclinic structure of greigite.

duction of the local Coulomb interaction accounted for by  $U_{\text{eff}}$ . Small values of  $U_{\text{eff}}$ , on the order of  $\sim 1$  eV, produce a dramatic improvement in the description of greigite, with the experimentally determined values for the lattice parameters and magnetic moments reproduced accurately.  $U_{\text{eff}}$  values greater than 3 eV produce solutions where the net magnetic moment is reduced by the occurrence of a magnetic moment on individual S atoms, antiparallel to that found on the octahedral Fe sites. This is accompanied by a decrease in the

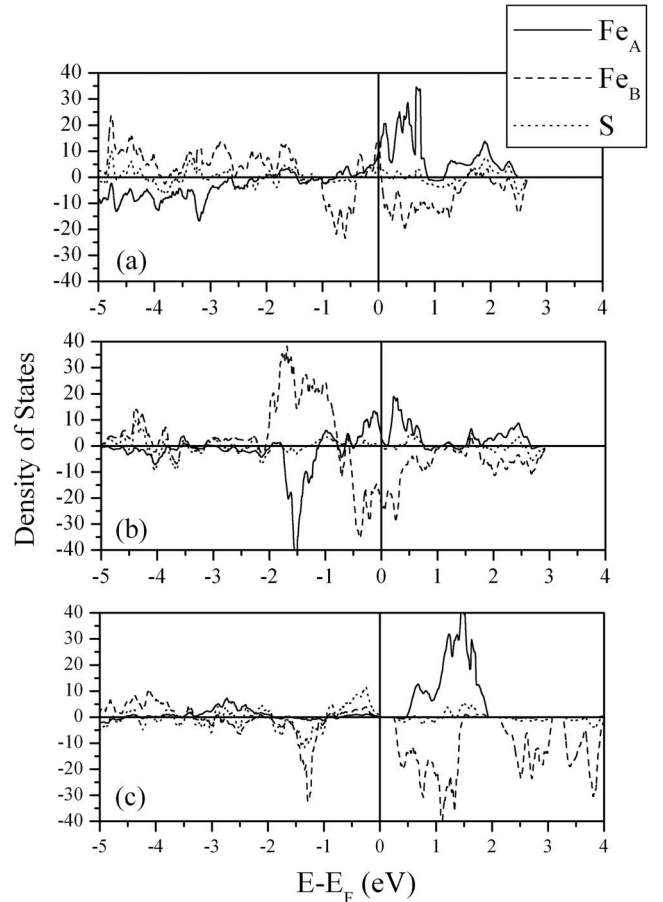


FIG. 5. DOS for the monoclinic form of greigite with (a)  $U_{\text{eff}} = 0$  eV, (b)  $U_{\text{eff}} = 1$  eV, and (c)  $U_{\text{eff}} = 5$  eV. Contributions from each of the sublattices are plotted.

number of electrons associated with the octahedral Fe atoms, which are transferred to the S atoms.

The two band schemes suggested by Spender *et al.*<sup>14</sup> for the electronic structure of greigite can now be reconsidered in the light of these results. The first scheme, where the octahedral Fe sites of greigite are occupied by a combination of ferric and ferrous iron is the scenario supported by our calculations for  $U_{\text{eff}} \leq 3$  eV. The second scheme, where the S ions reduce the ferric Fe ions so that all Fe in greigite is ferrous, is seen when  $U_{\text{eff}} \geq 3.5$  eV. It is not possible to discern the most correct value of  $U_{\text{eff}}$ , based only on the results presented here. However, since the experimental magnetization and the cell parameters are better reproduced at low  $U_{\text{eff}}$  values, we would suggest the use of  $U_{\text{eff}}=1$  eV for the GGA+U modeling of greigite. This value is lower than that suggested for magnetite<sup>36</sup> but very similar to that suggested for the iron sulphide troilite.<sup>34</sup> This is postulated to be due to the covalent nature of the Fe-S bond being more pronounced than that of the more ionic Fe-O bond, and thus the electron correlation represented by the  $U_{\text{eff}}$  parameter is weaker in the thiospinel. For  $U_{\text{eff}}=1$  eV the band structure calculations show greigite to be a semimetal, with the minority-spin band of the Fe octahedral sites providing charge carriers at the Fermi level. Further experimental investigations would be necessary in order to test this prediction.

Simulations of the theoretical monoclinic structure of greigite, based on the low-temperature magnetite structure, have shown that this form is not energetically favorable compared to the spinel structure for any  $U_{\text{eff}}$  values between 0 and 5 eV, indicating that greigite should not experience any Verwey-type transition to a monoclinic structure at low temperatures. While the mechanics of the Verwey transition are still an open area of research with many unanswered ques-

tions, previous *ab initio* calculations<sup>36</sup> have highlighted the importance of electron correlations in the transition, represented by a  $U_{\text{eff}}$  correction of around 3.2 eV or greater. Our calculations have shown that the stabilization of the monoclinic greigite structure with respect to the spinel would require unrealistically high values of  $U_{\text{eff}} > 5$  eV. Since it has been shown in this study that an accurate description of greigite is provided by a much lower  $U_{\text{eff}}$  value of 1 eV, it is postulated that the electron correlation associated with the Fe atoms in greigite is insufficient to facilitate a Verwey-type transition.

The finding that greigite is a ferrimagnetic semimetal, which conducts in only one spin polarization, places greigite within a very select group of materials with important applications in the field of spintronics,<sup>56</sup> which could be particularly relevant since iron sulfides offer scope for doping at other manipulations not possible in oxides.<sup>57</sup> In addition, greigite offers a much better example of a low-temperature iron spinel than magnetite, since it does not undergo a phase transformation at low temperature.

#### ACKNOWLEDGMENTS

Via our membership of the U.K.'s HPC Materials Chemistry Consortium, which is funded by EPSRC-GB (Contract No. EP/F067496), this work made use of the facilities of HECToR, the U.K.'s national high-performance computing service, which is provided by UoE HPCx Ltd at the University of Edinburgh, Cray Inc. and NAG Ltd, and funded by the Office of Science and Technology through EPSRC-GB's High End Computing Programme. We also thank the EPSRC-GB for financial support; the use of the Chemical Database Service at Daresbury.

\*a.devey@ucl.ac.uk

- <sup>1</sup>I. Letard, Ph. Sainctavit, N. Menguy, J.-P. Valet, A. Isambert, M. Dekkers, and A. Gloter, *Phys. Scr.*, **T115**, 489 (2005).
- <sup>2</sup>B. J. Skinner, R. C. Erd, and F. S. Grimaldi, *Am. Mineral.* **49**, 543 (1964).
- <sup>3</sup>A. P. Roberts and R. Weaver, *Earth Planet. Sci. Lett.* **231**, 263 (2005).
- <sup>4</sup>M. J. Dekkers, H. F. Passier, and M. A. A. Schoonen, *Geophys. J. Int.* **141**, 809 (2000).
- <sup>5</sup>M. J. Russell and A. J. Hall, *J. Geol. Soc. (London)* **154**, 377 (1997).
- <sup>6</sup>N. N. Nair, E. Schreiner, R. Pollet, V. Staemmler, and D. Marx, *J. Chem. Theory Comput.* **4**, 1174 (2008).
- <sup>7</sup>S. K. Goffredi, A. Warén, V. J. Orphan, C. L. Van Dover, and R. C. Vrijenhoek, *Appl. Environ. Microbiol.* **70**, 3082 (2004).
- <sup>8</sup>M. Pósfal, P. R. Buseck, D. A. Bazylinski, and R. B. Frankel, *Am. Mineral.* **83**, 1469 (1998).
- <sup>9</sup>A. R. Lennie, S. A. T. Redfern, P. E. Champness, C. P. Stoddart, P. F. Schofield, and D. J. Vaughan, *Am. Mineral.* **82**, 302 (1997).
- <sup>10</sup>A. R. Lennie, S. A. T. Redfern, P. F. Schofield, and D. J. Vaughan, *Miner. Mag.* **59**, 677 (1995).
- <sup>11</sup>Z. He, S.-H. Yu, X. Zhou, X. Li, and J. Qu, *Adv. Funct. Mater.* **16**, 1105 (2006).
- <sup>12</sup>W. Han and M. Gao, *Cryst. Growth Des.* **8**, 1023 (2008).
- <sup>13</sup>M. Uda, *Am. Mineral.* **50**, 1487 (1965).
- <sup>14</sup>M. R. Spender, J. M. D. Coey, and A. H. Morrish, *Can. J. Phys.* **50**, 2313 (1972).
- <sup>15</sup>L. Chang, A. Roberts, Y. Tang, B. D. Rainford, A. R. Muxworthy, and Q. Chen, *J. Geophys. Res.* **113**, B06104 (2008).
- <sup>16</sup>R. Aragón, *Phys. Rev. B* **46**, 5328 (1992).
- <sup>17</sup>D. J. Vaughan and M. S. Ridout, *J. Inorg. Nucl. Chem.* **33**, 741 (1971).
- <sup>18</sup>R. E. Vandenberghe, E. de Grave, P. M. A. De Bakker, M. Krs, and J. J. Hus, *Hyperfine Interact.* **68**, 309 (1991).
- <sup>19</sup>T. Zemčik and A. Cimbáliková, *Hyperfine Interact.* **83**, 499 (1994).
- <sup>20</sup>J. A. Morice, L. V. C. Rees, and D. T. Rickard, *J. Inorg. Nucl. Chem.* **31**, 3797 (1969).
- <sup>21</sup>M. Uhl and B. Siberchicot, *J. Phys.: Condens. Matter* **7**, 4227 (1995).
- <sup>22</sup>S. Sasaki, *Acta Crystallogr.* **B53**, 762 (1997).
- <sup>23</sup>J. M. D. Coey, M. R. Spender, and A. H. Morrish, *Solid State Commun.* **8**, 1605 (1970).
- <sup>24</sup>B. M. Moskowitz, R. B. Frankel, and D. A. Bazylinski, *Earth*

- Planet. Sci. Lett. **120**, 283 (1993).
- <sup>25</sup>A. P. Roberts, *Earth Planet. Sci. Lett.* **134**, 227 (1995).
- <sup>26</sup>V. I. Anisimov, F. Aryasetiawan, and A. I. Liechtenstein, *J. Phys.: Condens. Matter* **9**, 767 (1997).
- <sup>27</sup>H. P. Pinto and S. D. Elliot, *J. Phys.: Condens. Matter* **18**, 10427 (2006).
- <sup>28</sup>G. Kresse and J. Hafner, *Phys. Rev. B* **47**, 558 (1993).
- <sup>29</sup>G. Kresse and J. Hafner, *Phys. Rev. B* **49**, 14251 (1994).
- <sup>30</sup>G. Kresse and J. Furthmuller, *Comput. Mater. Sci.* **47**, 558 (1996).
- <sup>31</sup>G. Kresse and J. Furthmuller, *Phys. Rev. B* **54**, 11169 (1996).
- <sup>32</sup>M. C. Payne, M. P. Teter, D. C. Allan, T. A. Arias, and J. D. Joannopoulos, *Rev. Mod. Phys.* **64**, 1045 (1992).
- <sup>33</sup>D. Hobbs and J. Hafner, *J. Phys.: Condens. Matter* **11**, 8197 (1999).
- <sup>34</sup>A. Rohrbach, J. Hafner, and G. Kresse, *J. Phys.: Condens. Matter* **15**, 979 (2003).
- <sup>35</sup>A. Devey, R. Grau-Crespo, and N. H. de Leeuw, *J. Phys. Chem. C* **112**, 10960 (2008).
- <sup>36</sup>P. Piekarz, K. Parlinski, and A. M. Oles, *Phys. Rev. B* **76**, 165124 (2007).
- <sup>37</sup>L. Marsella and V. Fiorentini, *Phys. Rev. B* **69**, 172103 (2004).
- <sup>38</sup>R. Grau-Crespo, F. Corà, A. A. Sokol, N. H. de Leeuw, and C. R. A. Catlow, *Phys. Rev. B* **73**, 035116 (2006).
- <sup>39</sup>A. Walsh, S.-H. Wei, Y. Yan, M. M. Al-Jassim, J. A. Turner, M. Woodhouse, and B. A. Parkinson, *Phys. Rev. B* **76**, 165119 (2007).
- <sup>40</sup>J. P. Perdew, J. A. Chevary, S. H. Vosko, K. A. Jackson, M. R. Pederson, D. J. Singh, and C. Fiolhais, *Phys. Rev. B* **46**, 6671 (1992).
- <sup>41</sup>S. H. Vosko, L. Wilk, and M. Nusair, *Can. J. Phys.* **58**, 1200 (1980).
- <sup>42</sup>P. E. Blöchl, *Phys. Rev. B* **50**, 17953 (1994).
- <sup>43</sup>G. Kresse and D. Joubert, *Phys. Rev. B* **59**, 1758 (1999).
- <sup>44</sup>H. J. Monkhorst and J. D. Pack, *Phys. Rev. B* **13**, 5188 (1976).
- <sup>45</sup>N. D. Mermin, *Phys. Rev.* **137**, A1441 (1965).
- <sup>46</sup>F. Zhou, M. Cococcioni, K. Kang, and G. Ceder, *Electrochem. Commun.* **6**, 1144 (2004).
- <sup>47</sup>X. Du, Q. Li, H. Su, and J. Yang, *Phys. Rev. B* **74**, 233201 (2006).
- <sup>48</sup>G. Rollmann, P. Entel, A. Rohrbach, and J. Hafner, *Phase Transitions* **78**, 251 (2005).
- <sup>49</sup>A. I. Liechtenstein, V. I. Anisimov, and J. Zaanen, *Phys. Rev. B* **52**, R5467 (1995).
- <sup>50</sup>S. L. Dudarev, G. A. Botton, S. Y. Savrasov, C. J. Humphreys, and A. P. Sutton, *Phys. Rev. B* **57**, 1505 (1998).
- <sup>51</sup>I. V. Solovyev, P. H. Dederichs, and V. I. Anisimov, *Phys. Rev. B* **50**, 16861 (1994).
- <sup>52</sup>A. Sorkin, M. A. Iron, and D. G. Truhlar, *J. Chem. Theory Comput.* **4**, 307 (2008).
- <sup>53</sup>O. A. Vydrov, J. Heyd, A. V. Kruckau, and G. E. Scuseria, *J. Chem. Phys.* **125**, 074106 (2006).
- <sup>54</sup>R. F. W. Bader, M. T. Carroll, J. R. Cheeseman, and C. Chang, *J. Am. Chem. Soc.* **109**, 7968 (1987).
- <sup>55</sup>J. P. Wright, J. P. Attfield, and P. G. Radaelli, *Phys. Rev. B* **66**, 214422 (2002).
- <sup>56</sup>S. A. Wolf, D. D. Awschalom, R. A. Burhman, J. M. Daughton, S. von Molnár, M. L. Roukes, A. Y. Chtchelkanova, and D. M. Treger, *Science* **294**, 1488 (2001).
- <sup>57</sup>M. I. Katsnelson, V. Yu. Irkhin, L. Chioncel, A. I. Liechtenstein, and R. A. de Groot, *Rev. Mod. Phys.* **80**, 315 (2008).